

Preparation of Monodisperse and Highly Coercive $L1_0$ -FePt Nanoparticles Dispersible in Nonpolar Organic Solvents

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A method to prepare monodisperse and highly coercive $L1_0$ -FePt nanoparticles that are dispersible in nonpolar organic solvents such as toluene, chloroform, and hexane was developed. By vigorously stirring the SiO_2 -coated $L1_0$ -FePt nanoparticles synthesized by the “ SiO_2 -nanoreactor” method (*Appl. Phys. Lett.* **2005**, *87*, 032503) in a mixture of an aqueous NaOH solution, chloroform, and hexadecyltrimethylammonium bromide, the SiO_2 coating was dissolved off, and bare FePt nanoparticles could be extracted to the chloroform phase without degrading their magnetic properties. These particles showed a tendency to form a close-packed lattice on slow drying of the chloroform-based solution. The present success may promote the practical application to ultra-high-density magnetic recording and also may open the door to providing these particles with new physical and/or chemical functions.

Introduction

The FePt alloy with the $L1_0$ structure possesses a very high uniaxial magnetic anisotropy of ca. $6 \times 10^6 \text{ J/m}^3$, which is more than 10 times as high as that of the currently utilized CoCr-based alloys.¹ Superparamagnetic fluctuation of the room-temperature magnetization can thus be suppressed even for particles of 3 nm in diameter, making an appropriate array of these nanoparticles to be a promising candidate for future ultra-high-density magnetic recording media of $>1 \text{ Tbit/in.}^2$.^{2–9} The most basic requisite for the practical use would be the formation and the fixation of an array on a substrate

with the magnetic easy axis (i.e., the tetragonal c -axis) oriented normal to the substrate surface. Such a close-packed triangular array structure may be formed through self-organization under an external magnetic field if the $L1_0$ -FePt nanoparticles could be dispersed in a polymer binder. However, it has been experimentally difficult because of several drawbacks like insufficient Fe/Pt ordering, nonuniformity in particle size and shape, and lack of solubility in nonpolar organic solvents that dissolve a polymer binder.^{4c,5b,c}

Recently, we have succeeded in preparing monodisperse and highly coercive $L1_0$ -FePt nanoparticles that are dispersible in water by the strategic “ SiO_2 -nanoreactor” method.¹⁰ These particles are well-crystallized and have a room-temperature coercivity of 19 kOe in spite of their diameter

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- (9) Schematic illustration of a recording media with 2-D nanoparticle arrays, see <http://www.hitachigst.com/hdd/research/storage/pm/index.html>.

of only 6.5 nm. Also the particle moment could be highly oriented. Here, we report a method to render these $L1_0$ -FePt nanoparticles dispersible in nonpolar organic solvents (e.g., toluene, chloroform, and hexane). It means that we succeeded in removing the SiO_2 shell and extracting the $L1_0$ -FePt cores in these nonpolar organic solvents without degrading their high coercivity.

Experimental Section

Chemicals. Octyl ether (>95%, Tokyo Kasei), platinum(II) acetylacetonate (first grade, Wako Pure Chemicals), tetraethyl orthosilicate (special grade, Wako Pure Chemicals), oleic acid (first grade, Wako Pure Chemicals), oleyl amine (97%, Acros Organics), pentacarbonyl iron (99.99%, Aldrich), 1,2-hexadecandiol (90%, Aldrich), and hexadecyltrimethylammonium bromide (HTAB, GR grade, Nacalai Tesque) were used as received.

Preparation of the SiO_2 -Coated $L1_0$ -FePt Nanoparticles. The SiO_2 -coated $L1_0$ -FePt nanoparticles were prepared according to the procedures described elsewhere.^{10a} In brief, face-centered cubic (fcc) FePt nanoparticles with an average particle size of 6.4 nm were prepared first according to the method reported previously.^{3a} They were made water-soluble by encapsulation in HTAB using an oil-in-water microemulsion technique,¹¹ and their hydrophilic surface was subsequently coated by SiO_2 . The SiO_2 -coated fcc-FePt nanoparticles were then heated in a tube furnace (Lindberg, KTF045) at 900 °C for 1 h in a stream of H_2 (5%)/Ar(95%) for the transformation to the $L1_0$ structure. Previous work proved that this treatment provides well-crystallized and isolated $L1_0$ particles of 6.5 nm in diameter.^{10a,b}

Preparation of the $L1_0$ -FePt Nanoparticles Dispersible in Nonpolar Organic Solvents. The $L1_0$ -FePt nanoparticles dispersible in nonpolar organic solvents were prepared via an aqueous/organic biphasic reaction using HTAB as a phase-transfer reagent. In a typical run, finely ground SiO_2 -coated $L1_0$ -FePt nanoparticles (0.03 g), aqueous NaOH solution (4 M, 3 g), chloroform (5 g), and HTAB (0.5 g) were mixed and vigorously stirred for 24 h at room temperature. In this process the SiO_2 layer was dissolved off, and bare $L1_0$ -FePt nanoparticles were extracted to the chloroform phase. These were collected by centrifugation after adding ethanol to decrease the solubility and were again dispersed in chloroform containing oleic acid and oleyl amine. Undesirable precipitates were removed by centrifugation, and the $L1_0$ -FePt nanoparticles left in the supernatant were collected by centrifugation after adding ethanol. The thus-obtained powder was redispersed in chloroform containing oleic acid and oleyl amine and stored at room temperature. Typical yield of the $L1_0$ -FePt nanoparticles after the reaction is around 80%.

Characterization. Transmission electron microscopy (TEM) observations were performed using a Philips CM 200 TEM and a JEOL JEM-4000EX. TEM specimens were prepared by dropping the particle-containing original solution on a carbon-coated copper grid. The particle composition was determined by means of energy-dispersive X-ray analysis (EDX) on a JEOL JED 2140. Fourier transform infrared (FTIR) spectra were collected on a BioRad FTS-6000. The samples were mixed with KBr and compressed into pellets. Magnetic properties were measured with a SQUID mag-

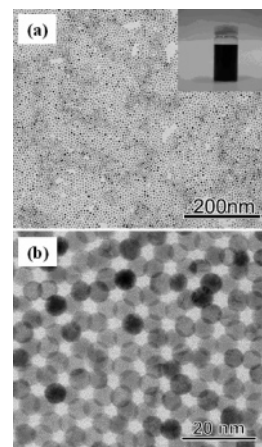


Figure 1. (a) Low-magnification TEM images of the $L1_0$ -FePt nanoparticles. (b) Tendency to form ordered superlattice structure. Inset shows an image of the chloroform solution containing the $L1_0$ -FePt nanoparticles.

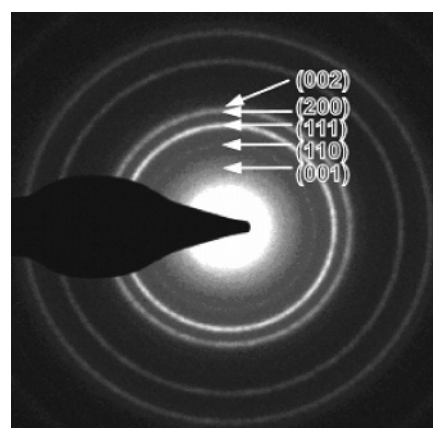


Figure 2. SAED pattern of the $L1_0$ -FePt nanoparticles.

netometer (Quantum Design MPMS-XL). For the EDX and FTIR measurements, the nanoparticles were precipitated from the solution by adding ethanol and then dried.

Results and Discussions

In the low-magnification TEM image in Figure 1a, we can see that the $L1_0$ -FePt nanoparticles are dispersed free of aggregation. The average particle size has been estimated to be 6.5 nm, which is the same as that of the SiO_2 -coated FePt particles. Their composition was determined to be $\text{Fe}_{55}\text{-Pt}_{45}$, which is again the same as that of the original fcc-nanoparticles. The inset shows a bottle of the chloroform solution of the $L1_0$ -FePt nanoparticles. Dispersability in other nonpolar organic solvents including toluene, hexane, and others was confirmed. By adding small amounts of oleic acid and oleyl amine to them, we could keep the solutions stable for months, free from any precipitation.

Thanks to their uniformity in size and also to their nearly spherical shape, the $L1_0$ -FePt nanoparticles tended to form two- and three-dimensional close-packed superlattices through spontaneous self-organization on slow drying as can be seen in Figure 1b. The inter-particle distance is ~ 10 nm, from which the thickness of the coating layer made of oleic acid and oleyl amine has been estimated to be ~ 1.8 nm.

Figure 2 shows a typical selective area electron diffraction (SAED) pattern. Note the appearance of the (110) ring

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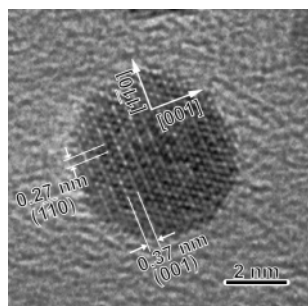


Figure 3. HRTEM image of the $L1_0$ -FePt nanoparticle.

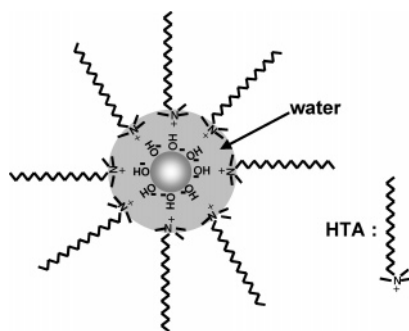


Figure 4. Schematic illustration of the double layer.

characteristic of the $L1_0$ -structure with $d_{110} = 0.27$ nm.¹² The high-resolution TEM (HRTEM) image in Figure 3 clearly shows that the $L1_0$ -FePt nanoparticles are single crystalline with the (110)-superlattice contrast clearly seen up to the surface end. The particle is faceted on $\{111\}$ and $\{011\}$ planes.

These solutions could dissolve a variety of polymers such as poly(methyl methacrylate) and polystyrene, from which thin polymer films containing uniformly dispersed $L1_0$ -FePt nanoparticles could be easily formed on various kinds of substrates. Their chemical and physical properties are currently under investigation.

Here, we discuss various steps of the extraction process. The first step is the dissolution of the SiO_2 coating. It was amorphous in nature even after the treatment at 900 °C in flowing H_2 (5%)/Ar(95%) atmosphere as found in our previous work.^{10a} As well-known SiO_2 depolymerizes under strong basic conditions.¹³ In the previous work, we dissolved it just by suspending in a tetramethylammonium hydroxide solution (10 wt % aqueous solution), while in the present process basicity was provided by the aqueous NaOH solution. The bared $L1_0$ -FePt nanoparticles thus obtained were then extracted to the chloroform phase with the aid of HTAB. The mechanism contains most probably the formation of a double layer such as schematically drawn in Figure 4, which consists of negatively charged hydroxide ions adsorbed to the surface of the $L1_0$ -FePt nanoparticles and countercations of hexadecyltrimethylammonium (HTA). A small amount of water might also be contained as suggested in the same figure. The FTIR spectrum of the $L1_0$ -FePt nanoparticles just after the extraction to the chloroform phase has peaks

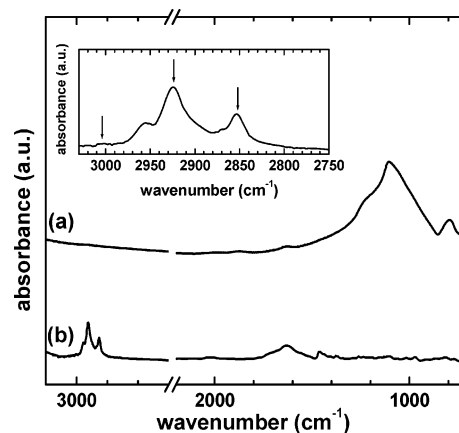


Figure 5. FTIR spectra of the $L1_0$ -FePt nanoparticles at the various preparative steps: (a) the SiO_2 -coated state and (b) after the purification step. Inset shows the FTIR spectrum of the $L1_0$ -FePt nanoparticles after the purification step in the 2750–3030 cm^{-1} region. The arrows indicate the characteristic absorption peaks of the oleyl group.¹⁵

assigned to HTA (Figure S1 and Figures S2(B) in Supporting Information), which supports this double-layer view. These $L1_0$ -FePt nanoparticles could favor being in the chloroform phase rather than in the aqueous phase owing to the hydrophobic character of the long alkyl chains of HTA. In the extraction process, the countercations are supposed to play two important roles: one is to form the double layer while in the aqueous phase,¹⁴ and the other is to promote dispersion in the chloroform phase. In other words, they should have a suitable degree of amphiphilicity for successful extraction. Actually, the use of a more hydrophobic cation like trioctylmethylammonium in place of HTA and also the use of a more hydrophilic cation of tetramethylammonium resulted in very limited success. The moderately hydrophobic and hydrophilic character of HTA must have been the key to the successful extraction.

Figure 5 shows FTIR spectra of certain samples. The spectrum of the SiO_2 -coated $L1_0$ -FePt nanoparticles (Figure 5a and Figure S2(A) in Supporting Information) consists of essentially two peaks around 1100 and 800 cm^{-1} assigned to amorphous SiO_2 . On the other hand, the spectrum of the $L1_0$ -FePt nanoparticles after the purification process (Figure 5b and Figure S2(C) in the Supporting Information) has no peaks of SiO_2 origin but has new peaks arising from the organic stabilizers. The inset shows the spectrum magnified in the 2750–3030 cm^{-1} region. The observed peaks are assigned to the oleyl group,¹⁵ though the spectrum taken just after the extraction to the chloroform phase showed peaks from HTA. Replacement of HTA by oleic acid and oleyl amine must have taken place in the subsequent purification steps (Figure S2 in Supporting Information).

Magnetic properties of dried and chloroform-dispersed $L1_0$ -FePt nanoparticles studied through magnetization measurements using a SQUID magnetometer were essentially the same as those reported in the previous paper.¹⁰ Shown

(12) The (001) superlattice ring could not be seen due to the diffraction from amorphous carbon. The (002) superlattice ring and the (202) ring are too close to each other to be separated.

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(15) The peaks at 2854 and 2924 cm^{-1} are assigned to the symmetric and asymmetric CH_2 stretching modes, and the small peak at 3006 cm^{-1} is due to the $\nu(\text{C}-\text{H})$ mode of the $\text{C}-\text{H}$ bond adjacent to the $\text{C}=\text{C}$ bond.

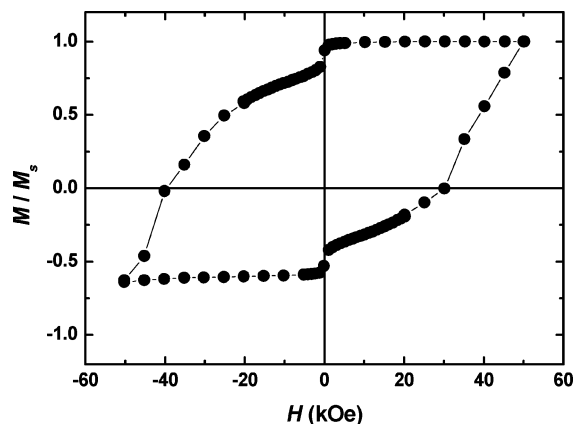


Figure 6. Hysteresis loop of the chloroform-dispersed $L1_0$ -FePt nanoparticles measured at 150 K after cooling in an applied magnetic field of 10 kOe.

in Figure 6, for example, is the hysteresis loop of the chloroform-dispersed nanoparticles measured after field-cooling down to 150 K at 10 kOe.¹⁶ At this temperature, the matrix solution was frozen, any motion of the FePt nanoparticles thus being suppressed. Here, M_s represents the saturation magnetization.¹⁷ It is notable that the remanent magnetization (M_r : magnetization at $H = 0$ kOe) is almost equal to the saturation magnetization as expected from an anisotropic hard magnet. On reversing the applied field, the magnetization shows a small and quick drop to $M/M_s = 0.83$ and then decreases very slowly down to $M/M_s = -0.63$, not to -1 as expected. This indicates that the reversal of magnetization did not finish completely because the maximum field available was not large enough in comparison with the coercivity of ~ 40 kOe. The initial small drop

(16) The loop starts from $(M/M_s, H) = (1, 50 \text{ kOe})$, reaches the point of $(M/M_s, H) = (-0.63, -50 \text{ kOe})$, and comes back to the starting point.

(17) Saturation magnetization is defined as the magnetization at $H = 50$ kOe.

mentioned above is attributed to the presence of a small amount of untransformed fcc-FePt nanoparticles as suggested from Mössbauer spectroscopic measurements on the SiO_2 -coated samples.^{10c} We note here that it is beyond the scope of the present work to measure the degree of the alignment of particle moment as a function of temperature, applied field, and particle size for comparison with recent theoretical calculations.^{5c,18}

Conclusions

We have developed a method to prepare monodisperse and magnetically highly coercive $L1_0$ -FePt nanoparticles dispersible in nonpolar organic solvents like toluene, chloroform, and hexane. We believe that the solubility in nonpolar organic solvents will allow us to attach new functionalities to the surface of the $L1_0$ -FePt nanoparticles through a variety of organic and organometallic chemical reactions. Such $L1_0$ -FePt nanoparticles will be of great potentiality not only for ultra-high-density magnetic recording but also for magnetochemical and magnetophysical surface functionalities.

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Supporting Information Available: FTIR spectrum of the $L1_0$ -FePt nanoparticles just after the extraction to the chloroform phase (Figure S1) and schematic representation of the extraction and purification processes (Figure S2). This material is available free of charge via the Internet at <http://pubs.acs.org>.

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